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### Crystalline Carbon Nitride for Photocatalytic Phenol Degradation: Effect of Precursor and Salt Melt Amounts

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Abstract. Development of visible-light-active photocatalyst is an important approach to utilize solar energy in the future. The attempts to improve the crystallinity of photocatalyst have been the focus of the research. Recently, the use of potassium chloride-lithium chloride (KCl-LiCl) salt melt has been reported to increase the crystallinity of carbon nitride. In this work, precursor and salt melt amounts were varied to study the properties and the photocatalytic activity of the crystalline carbon nitride for phenol degradation. When the precursor amount was too low (1 g), no product was obtained due to the decomposition of precursor. When it was too high (4 g), the product showed the characters of amorphous CN. Optical properties also showed that when the precursor amount was 3–4 g, the incomplete condensation process occurred and defects were formed due to the excessive washing. The specific surface area also decreased with the increase of the precursor amount from 2 to 4. When the amount of salt melt was too low (2.5 g), the crystalline CN could not be obtained. However, when the amount of salt melt was increased to 7.5 g, impurities could not be avoided. Longer time and multiple washing processes were required in order to remove the impurities, which certainly affected the crystallinity. Photocatalytic activity test showed that the high activity was obtained on the sample prepared using the optimum amount of precursor (2 g) and salt melt (5 g), which would be due to the high crystallinity and large specific surface area.

#### INTRODUCTION

Phenol is known to be a toxic molecule and the permission exposure level in bottle water has been set to 0.001 mg/L by the Food and Drug Administration (FDA), while the Occupational Safety and Health Administration (OSHA) has set the legal limit of phenol in air to 5 ppm over an 8-hour work day [1]. While various methods have been reported to treat phenol, one promising approach that is worth to be explored is by photocatalytic reaction. The photocatalytic degradation of phenol has been shown to provide the environmentally safe pathway and the process in principle only needs the light energy and suitable semiconductor photocatalyst [2-4].

In order to fully utilize the abundant solar light, the photocatalyst shall be able to absorb the visible light which is larger part than the ultra violet (UV) region in the solar spectrum. Carbon nitride has been recognized as an organic semiconductor that could work as a potential heterogeneous photocatalyst under visible light irradiation [5-7]. While many methods have been developed to synthesize carbon nitride, most of them are in amorphous form. However,

amorphous carbon nitride usually shows less photocatalytic activity than the crystalline carbon nitride, either for photocatalytic hydrogen production [8] or photocatalytic degradation of organic pollutant such as phenol [9]. Therefore, the development of methods to synthesize crystalline carbon nitride is still an emerging research in this photocatalytic field. Recently, our group reported the use of potassium chloride-lithium chloride (KCl-LiCl) salt melt to synthesize crystalline carbon nitride [9]. In this study, we further investigated the effects of precursor amount and salt melt amount on the properties and photocatalytic activity of the obtained carbon nitride.

#### **EXPERIMENTAL**

The crystalline carbon nitride samples were prepared in the similar way to the previous report [9]. The precursor of carbon nitride was prepared by calcining a certain amount of urea (CH<sub>4</sub>N<sub>2</sub>O, QRëC, 99%, 30 g) at 350 °C for 6 hours. In order to investigate the effect of the precursor amount, 2–4 g of precursor was added to 30 mL of ethanol solvent, followed by continuous stirring at 100 °C until all ethanol solvent was successfully evaporated. The obtained solid precursor was transferred to a porcelain mortar. A certain amount of salt melt (5 g), which was potassium chloride (KCl, Fisher Chemicals, 99.5%, 2.74 g) and lithium chloride (LiCl, Sigma-Aldrich, 99%, 2.26 g), was added into the mortar and the mixture was then ground together. The mixture was then heated at a ramp of 2.2 °C/min to reach 550 °C in a muffle furnace. The heating at 550 °C was held for 4 hours before cooling it to room temperature. The final product was ground to powder and the washing process was performed to remove residual salts with boiling water. The final products were labeled as x–CN, which x refers to 2, 3, and 4 g. To study the effect of the salt melt amount, a similar synthesis method was performed. The amounts of precursor (2 g), reaction temperature (550 °C), and reaction time (4 hours) were fixed, while the amount of salt melt was varied to 2.5 g (1.37 g KCl, 1.13 g LiCl), 5 g (2.74 g KCl, 2.26 g LiCl), and 7.5 g (4.11 g KCl, 3.39 LiCl). The final products were labeled as CN–x, which x refers to 2.5, 5, and 7.5 g of salt melt.

All the samples were further characterized by X-ray diffractometer (XRD, Bruker D8 Advance), diffuse reflectance ultraviolet-visible (DR UV-vis, Shimadzu UV-2600), Fourier transform infrared spectrophotometer (FTIR, Nicolet-iS50), and nitrogen adsorption-desorption isotherm analyzer (Quantachrome NOVAtouch LX4). The XRD patterns were measured with  $CuK_{\alpha}$  irradiation ( $\lambda$  = 1.5406 Å). As the reference for the measurement of DR UV-vis spectra, barium sulphate (BaSO<sub>4</sub>) was used. The technique used in the FTIR spectra measurement was potassium bromide (KBr) pellets technique. Before the measurement, the sample was ground in the mortar and mixed with KBr. The sample ratio was fixed to 1:100 and it was compressed under pressure below than 2 tonnes for 5 minutes to obtain a self-supporting pellet. All the characterization measurements were carried out at room temperature, except for the nitrogen adsorption-desorption isotherm profiles, which were recorded at -196.15 °C (77 K). Prior to the measurement of the isotherm profiles, the samples were dried at 100 °C. For degassing part, the samples were heated at 180 °C for 3 hours at the ramp of 10 °C/min. The specific surface area (SA) was determined by the Brunauer-Emmett-Teller (BET) approach from the adsorption part, while the pore diameter (PD) and pore volume (PV) were derived by the equation from Barret-Joyner-Halenda (BJH).

Phenol (C6H6O, Scharlau Chemie, 99.5%) was used as the model of organic pollutant. The photocatalytic degradation reaction was performed in the similar way to the previous work [9]. Photocatalyst (50 mg) was first dispersed in 50 ml of 50 ppm phenol. Prior to the photocatalytic reaction, the solution was stirred continuously in the dark condition for 30 mins in order to reach the equilibrium of adsorption-desorption. The photocatalytic reaction was carried out under solar light irradiation (150 W,  $\lambda > 230$  nm, I = 110, 000 Lux) for 6 hours. After completion, the solution was filtered to separate the sample. The final remaining concentration of the solution was determined by using a high-performance liquid chromatography (HPLC, Shimadzu LC-20AT) equipped with a UV detector. The final concentration of phenol was determined and the percentage of phenol degradation was calculated from the ratio of degraded phenol to its initial concentration.

#### RESULTS AND DISCUSSION

#### **Effect of Precursor Amount**

The effect of the precursor amount in the preparation of crystalline CN was studied, where the amount of precursor was varied at 2, 3 and 4 g. In the preliminary study, 1 g of the precursor was also investigated. However, the final product was believed to be fully decomposed as the amount of precursor was too low. The structural properties of the synthesized CN samples were characterized by XRD. Shown in Figure 1 are the XRD patterns of 2–CN, 3–CN, and

4–CN. From the diffractograms, only the 2–CN and 3–CN samples showed the presence of crystalline phase with 3–CN showed a decreased intensity. For the 2–CN sample, the crystalline peaks were observed at diffraction peaks at  $2\theta$  of 20.7 (110), 24.5 (200), 29.3 (102) and 32.4° (210). Meanwhile, the interlayer distance of graphite-like of CN and in-plane arrangement of nitrogen-linked heptazine CN heterocycle can be observed at  $2\theta$  of ca. 26.5 (002) and 11.8° (100), which gave d-spacing values of 0.33 and 0.74 nm, respectively. For the 3–CN sample, the crystalline peaks were observed at  $2\theta$  of 21.1 (110), 24.6 (200), 29.4 (102) and 32.1° (210). The broad and intense peaks at 27.1 (002) and 12.3° (100) were corresponded to an interlayer distance of CN and to the in-plane arrangement of CN heterocycle, which gave d-spacing values of 0.33 and 0.72 nm. For the 4–CN, no crystalline peak was observed but the diffraction peaks at  $2\theta$  of 27.2 and 11.9° could be detected. These peaks corresponded to the interlayer distance of graphite-like of CN and in-plane arrangement of nitrogen-linked triazine CN heterocycle, which gave d-spacing values of 0.33 and 0.74 nm, respectively. The XRD patterns of samples with crystalline [10-13] and amorphous phase [14-18] were in good agreement to the previous studies when the CN was synthesized by urea or other organic nitrogen-rich compounds such as melamine, dicyandiamide and cyanamide.

As presented in Fig. 1(b), increasing the amount of precursor to 3 g caused the decrease in crystallinity as well as induced back to the amorphous phase when the amount of 4 g was used (Fig. 1(c)). This result suggested that 2 g was the optimum amount of precursor needed. The decrease in the crystallinity observed on the 3–CN sample would be due to the insufficient amount of salt melts (ionic solvent) when the amount of precursor was increased. The (002) peak was also slightly shifted to the larger angle, suggesting the decrease in the interlayer distance between the CN graphitic layer. The shifting to the larger angle could be related to the decrease in crystallite size owing to the decrease in crystallinity. In addition to the crystallinity, it was reported that the (002) peak for poly(triazine imide) lies at ca. 26.5° [10,11,19], while heptazine—base unit has a diffraction peak of (002) plane at  $2\theta$  of 27.1° [11,19]. Therefore, the shifting to larger angle might also an indication for the presence of heptazine-based unit or tri-s-triazine rather than poly(triazine imide) or triazine [19].

Since the presence of heptazine-based unit or tri-s-triazine suggested the presence of amorphous phase, it can be suggested that the 3–CN sample consisted of a more amorphous phase and reduced crystalline phase than the 2–CN. On the other hand, the 4–CN sample showed only the amorphous phase. The decreased intensity on (002) plane might be due to excessive washing process during the removal of excess KCl-LiCl salt melt. The excessive washing process involved repeated and multiple steps of washing of the sample with distilled and boiling water with continuous stirring up to 2 hours. This procedure could affect the structural order of CN. This result suggested the need for optimum amount of precursor in order to maintain the crystallinity of the CN.

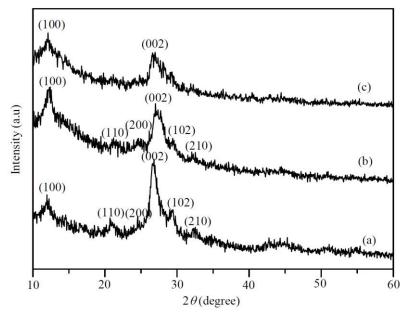


FIGURE 1. XRD patterns of (a) 2-CN (b) 3-CN and (c) 4-CN samples

The absorption properties of all the prepared CN photocatalysts were analysed by DR UV-Vis spectrophotometer. Figure 2(a) revealed the absorption spectra of 2–CN, 3–CN, and 4–CN. From the DR UV-Vis spectra, it was demonstrated that the higher the crystallinity, the higher the light absorption. For the 2–CN and the 3–CN samples, there were three main peaks were observed at the wavelength of 275, 320 and 373 nm. On the other hand, two major peaks were observed for the 4–CN sample, which appeared at the wavelength of 291 and 335 nm. The absorption peaks at 275 and 373 nm were likely to be attributed to the transition of  $\pi \to \pi^*$  and  $\pi \to \pi^*$  due to the bonds of C=N and C–N, respectively, owing to the electronic transition of aromatic heptazine/poly(triazine imide) units. The peak at ca. 320 nm corresponded to the transition of  $\pi \to \pi^*$  due to the existence of C=O bond [20-22]. The formation of C=O in CN was likely due to the incomplete condensation of precursor during the polymerization process [15]. It was demonstrated that the peak intensity at 320 nm corresponded to C=O functional group on the 3–CN sample was increased, suggesting the increase in the incomplete condensation process.

All the samples showed a different edge of absorption to each other. The 2–CN sample showed the absorption up to 460 nm, followed by the 3–CN with absorption up to 450 nm. The 4–CN showed the lowest absorption range with up to 400 nm, but demonstrated a slight increase of absorption intensity at *ca.* 420–600 nm. The small upshift absorption could be caused by the formation of defects resulting from the excessive washing of the samples during the removal of excess salt melts. The excessive washing might result in a disturbance of the CN network, leading to the formation of more defects in the structure. It was reported that such defects would scatter light that resulted in a better reflection of light [14, 23].

The optical band gap energy values of the prepared samples were estimated via Tauc plot by plotting the  $(\alpha hv)^{1/2}$  versus hv and the plot was presented Fig. 2(b). The Eg values of the respective samples are shown as the values in the x axis obtained from the linear extrapolation. The Eg values were altered when increasing the amount of the precursor. The Eg values were estimated to be 2.82, 2.96 and 3.19 eV for 2–CN, 3–CN and 4–CN, respectively. The increased band gap energy on the 4–CN sample was resulted from the formation of more defects and destroyed structure. These values suggested the need of optimum amount of precursor to prepare the crystalline CN in order to maintain the band gap energy value to enable them acting as visible light-driven photocatalysts.

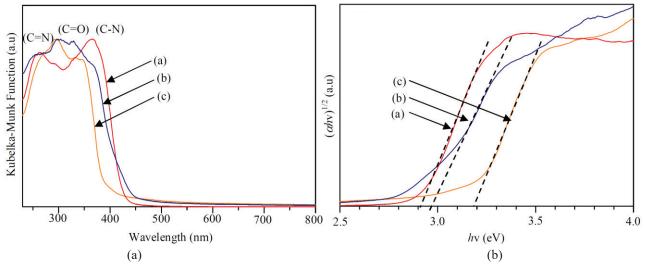


FIGURE 2. (a) DR UV-Vis spectra and (b) Tauc plots of (a) 2-CN (b) 3-CN and (c) 4-CN samples

The characteristic of the chemical bonding and functional groups of the prepared samples were investigated by FTIR spectrophotometer. Figure 3 shows the FTIR spectra of 2–CN, 3–CN and 4–CN samples. From the spectra, the formation of CN heterocycles or triazine units could be observed from the vibration band of all samples at the region of ca. 800 and 1200–1700 cm<sup>-1</sup>. The region at ca. 800 cm<sup>-1</sup> was assigned to bending mode of out-of-plane triazine units, while the multiple bands at ca. 1200–1700 cm<sup>-1</sup> were corresponded to stretching mode of triazine units and tris-triazine units [10-12,14,15]. On the other hand, the small intense peak presented at ca. 2170 cm<sup>-1</sup> observed in all samples indicated the formation of C $\equiv$ N and N=C=N. The formation of both C $\equiv$ N and N=C=N were typical for CN with crystalline materials due to intercalation of lithium and chloride ions in CN networks during the crystallization

process. It also implied that the continuity of CN network was disturbed [10-12]. Although the 4–CN sample was amorphous, it also showed the presence of both C $\equiv$ N and N=C=N, suggesting that the continuous structure or polymer of CN was broken. The broken structure might be related to the excessive washing during removal of excess salt melts in the sample. The functional group O–H at region ca. 3000–3800 cm $^{-1}$  indicated the adsorbed water molecules in all the CN samples.

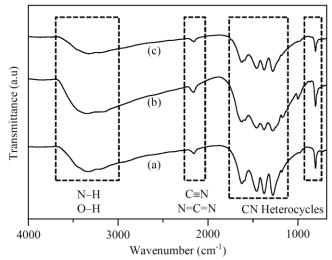


FIGURE 3. FTIR spectra of (a) 2–CN (b) 3-CN and (c) 4-CN samples

Figure 4 shows the isotherm profiles and BJH pore size distribution (shown in the inset) of the 2–CN, 3–CN and 4–CN. Based on the isotherms, the 2–CN and the 3–CN have type IV isotherm with H3 hysteresis according to IUPAC classification, suggesting the presence of the porous structure. However, the 4–CN showed a type III isotherm with no hysteresis loop. The BET specific surface area values for 2–CN, 3–CN and 4–CN were 73, 55 and 15 m<sup>2</sup> g<sup>-1</sup>, respectively. The pore size distribution for each sample based on the increasing amount of precursors was 3.62, 3.63 and *ca.* 12 nm, respectively. It is noted that the 4–CN sample showed a broad pore distribution, where the presence of pores might be due to the formation of cavities resulting from the releases of gases (NH<sub>3</sub>) during the polymerization process as also reported elsewhere [17]. The decrease in the BET specific surface area values might be due to the decrease in crystallinity [11]. It would be also due to the possibility of the broken structure of CN networks as can be seen on the 4–CN, which showed a big drop in the BET specific surface area to 15 m<sup>2</sup> g<sup>-1</sup>. The reason for the similar pore size distribution values for both 2–CN and 3–CN were due to the fixed amount of salt melts added to the samples as the formation of porous was resulted from the presence of void channels due to the ABA stacking of crystalline CN [13].

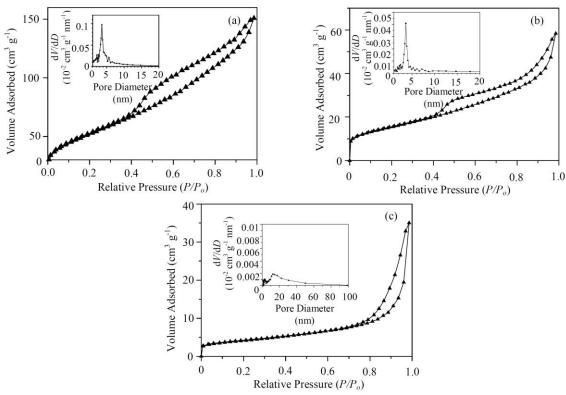
The photocatalytic performance of the prepared CN samples was tested for the degradation of phenol under solar light irradiation for 6 hours. The percentages of the photocatalytic degradation along with the physical and chemical properties of the samples are listed in Table 1. As presented in Table 1, the 2–CN showed the highest activity compared to others with 24% of phenol degradation after 6 hours under solar light irradiation, while the 3–CN and the 4–CN only showed 12 and 5% degradation, respectively. Based on the physical and chemical properties presented in Table 1, it was obvious that the high photocatalytic activity achieved on the 2–CN was due to its high crystallinity and large BET specific surface area as crystallinity could provide an efficient electron charge transfer and high surface area could give many active sites for photocatalytic oxidation to occur. For the 3–CN, the decreased photocatalytic activity to almost half of that on the 2–CN was due to decreased crystallinity and BET specific surface area. However, the band gap energy did not much affect the photocatalytic activity as its value was almost close to that of the 2–CN. On the other hand, the low photocatalytic activity obtained on the 4–CN was clearly due to its amorphous phase, large band gap energy and low surface area. These results suggested the importance of crystallinity, high surface area and low band gap energy to prevent the degradation percentage from dropping to a low value.

TABLE 1. Properties and percentage of phenol degradation after 6 hours under solar light irradiation for g-CN samples

Sample	Crystallinity	$E_{\rm g}^{\rm a}$ (eV)	$S.A^b (m^2 g^{-1})$	Degradation (%)
2–CN	✓	2.82	73	24
3-CN	✓	2.96	55	12
4–CN	×	3.19	15	5

<sup>&</sup>lt;sup>a</sup>Band gap energy values were determined via Tauc plot.

<sup>&</sup>lt;sup>b</sup>Specific surface areas were determined via BET technique.



**FIGURE 4.** Isotherm profiles and their respective BJH pore size distribution curves (inset) of (a) 2–CN and (b) 3–CN and (c) 4–CN samples.

#### **Effect of Salt Melts Amount**

The structural properties of the synthesized CN–x photocatalysts were characterized by XRD. Based on the XRD patterns in Fig. 5, all samples showed the (002) and (100) planes that corresponded to the interlayer distance of graphite-like CN and in-plane arrangement of poly(triazine imide) and heptazine-based CN unit. Both planes could be observed at the diffraction angle of *ca.* 26.8 and 11.9°. No other peak was observed on the CN–2.5 sample (Fig. 5a). In contrast, it was observed that the CN–5 and the CN–7.5 samples showed the presence of crystalline properties. As revealed in Fig. 5(b) and 5(c), the (110), (200), (102) and (210) planes at *ca.* 20.4. 24.6, 29.3 and 32.1° were corresponded to crystalline peaks [10-18]. The CN–7.5 sample also showed the presence of unfavoured impurities (Fig. 5(c)). The impurities are coming from the excess salt melt that was difficult to be removed when the amount was too high. As outlined in the experimental part, all the samples were washed with boiling water with the stirring process of 1–2 hours. The CN–7.5 sample was subjected to second wash with a similar approach but the process reduced the crystallinity. From these XRD patterns, it can be proposed that 5 g was the optimum amount of salt melt required to synthesize crystalline CN with the best crystalline phase.

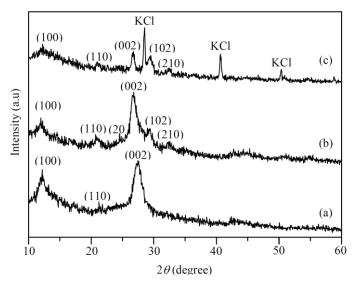


FIGURE 5. XRD patterns of (a) CN-2.5 (b) CN-5 and (c) CN-7.5 samples

The absorption properties of all the prepared CN photocatalysts were analysed by DR UV-Vis spectrophotometer. The absorption spectra of the CN-2.5, the CN-5, and the CN-7.5 are shown in Fig. 6(a). All samples showed three main peaks that corresponded to C=N, C=O and C-N owing to transition of  $\pi\to\pi^+$ ,  $n\to\pi$  and  $\pi\to\pi^*$  and  $\pi\to\pi^*$ , respectively [11,20,24,25]. The presence of C=O was related to the formation of defect owing to the less condense product when using the urea-based precursor. Based on the absorption pattern, it can be also noted that increasing the amount of salt melt from 2.5 to 5 g led to the increase in absorption towards longer wavelength. However, a further increase to 7.5 g did not give a further increment of absorption edge wavelength. The band gap energy values of prepared samples were estimated via Tauc plot as presented in Fig. 6(b). The obtained band gap energy values for CN-2.5, CN-5 and CN-7.5 were 2.98, 2.87 and 3.0 eV, respectively. The CN-7.5 sample showed an increase in its band gap energy value, which may be caused by the slight decrease in the layer thickness due to insertion of more Li<sup>+</sup> and Cl<sup>-</sup> in the layer and the presence of excess salt melt [26].

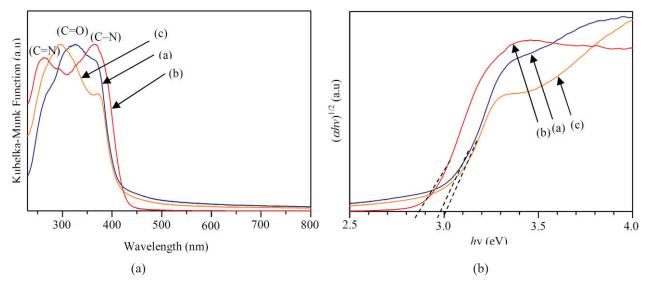


FIGURE 6. [A] DR UV-Vis spectra and [B] Tauc plots of (a) CN-2.5 (b) CN-5 and (c) CN-7.5 samples.

The characteristic of the functional groups of the samples was investigated by FTIR spectrometer. Figure 7 shows the FTIR spectra of the CN-2.5, the CN-5, and the CN-7.5. From the spectra, it can be observed that there was no significant change in the functional groups when the amount of salt melt was varied. The successful formation of CN

network can be observed from the peaks at the region of 810 and 1200–1700 cm<sup>-1</sup>, which attributed to CN heterocycles. The band at 810 cm<sup>-1</sup> could be assigned to the bending mode of out-of-plane heptazine-based and/or poly(triazine imide) units. Meanwhile, the bands at ca. 1200–1700 cm<sup>-1</sup> were corresponded to the stretching mode of heptazine and/or poly(triazine imide) units. In addition, the overlapping bands of N–H and O–H groups could be observed as the broad and wide bands at region of ca. 3000–3700 cm<sup>-1</sup> [10-2,14-17,26]. The small and intense single peak appeared at region ca. 2170 cm<sup>-1</sup> was due to the broke of CN continuity framework [10-12, 14, 15], owing to the intercalation by both Li<sup>+</sup> and Cl<sup>-</sup> ions in the crystalline CN. However, it can be noted that the peak attributed to C $\equiv$ N and N=C=N at 2170 cm<sup>-1</sup> for CN–2.5 sample (Fig. 7(a)) was less intense compared to others (Fig. 7(b) and (c)) due to the less amount of salt melts added.

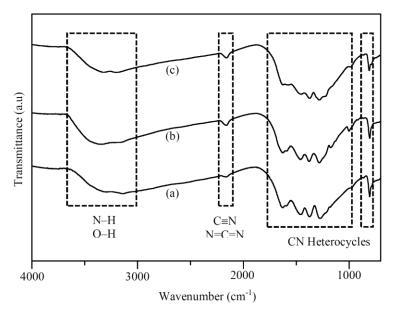
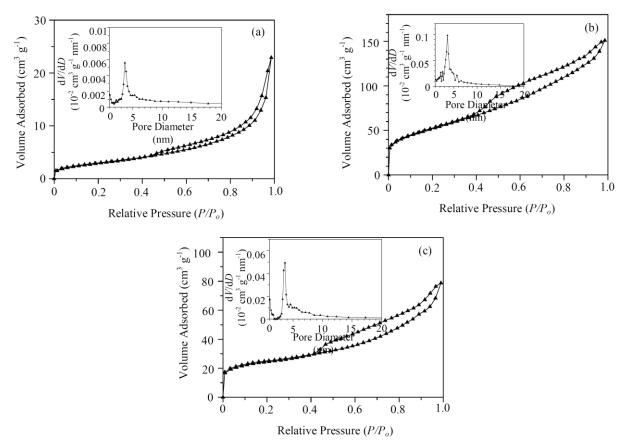


FIGURE 7. FTIR spectra of (a) CN-2.5, (b) CN-5, and (c) CN-7.5 samples.

Figure 8 reveals the isotherm profiles of nitrogen adsorption-desorption and BJH pore size distribution (inset) of the CN–2.5, the CN–5, and the CN–7.5. All samples showed the type IV isotherm with type H3 hysteresis loop, suggesting the formation of the porous structure. The values of the BET specific surface area for the CN–2.5, the CN–5, and the CN–7.5 were 60, 73, and 11 m<sup>2</sup> g<sup>-1</sup>, respectively. From the insets of Fig. 8, the BJH pore size distribution analysis showed that the CN–2.5, the CN–5, and the CN–7.5 samples have an average pore size of 3.64, 3.62 and 3.63 nm, respectively. Meanwhile, the average pore volume for the CN–2.5, the CN–5, and the CN–7.5 were 0.03, 0.23 and 0.11 cc g<sup>-1</sup>.

From the obtained BET specific surface area values, it could be demonstrated that the specific surface area of samples increased as increasing the salt melt amount from 2.5 to 5 g. The large surface area can be related to the high crystallinity phase of the prepared samples. However, increasing addition of eutectic mixture to 7.5 g caused a decrease in the specific surface area down to 11 from 73 m<sup>2</sup> g<sup>-1</sup>. The decrease in the specific surface area of the CN–7.5 sample was due to the decrease in the crystallinity and the presence of salt melts that affected the measurement of the surface area. In addition, the decreased surface area could be associated to the destroyed CN framework due to the excessive washing of the sample. From this study, it was demonstrated that that the large specific surface area could be obtained by increasing the crystallinity. The larger amount of salt melt caused further washing process which could result in the lower crystallinity and lower specific surface area.

Table 2 shows the percentage of phenol degradation of the prepared CN-g samples after 6 hours of irradiation. From the Table, the CN-2.5, the CN-5, and the CN-7.5 showed percentage degradation of 14, 24 and 11%, respectively. It could be clearly observed that the high photocatalytic activity on the CN-5 was due to its high crystallinity, improved specific surface area and low band gap energy. The lower activities obtained on the CN-2.5 and CN-7.5 would be caused by the lower crystallinity, the presence of impurities, lower specific surface area, and higher band gap energy. These results again demonstrated that the high crystallinity, large specific surface area and low band gap energy are the key properties that lead to good photocatalytic activity.



**FIGURE 8.** Isotherm profiles and their respective BJH pore size distribution curves (inset) of (a) CN–2.5, (b) CN–5, and (c) CN–7.5 samples.

TABLE 2. Properties and percentage of phenol degradation after 6 hours under solar simulator irradiation for CN-g samples

Sample	Crystallinity	Ega (eV)	$S.A^{b} (m^{2} g^{-1})$	Degradation (%)
CN-2.5	×	2.98	60	14
CN-5	✓	2.82	73	24
CN-7.5	✓	3.0	11	11

<sup>&</sup>lt;sup>a</sup>Band gap energy values were determined via Tauc plot.

#### **CONCLUSION**

Amounts of precursor and salt melt affected the physicochemical properties and photocatalytic performance of the crystalline carbon nitride. When the amount of the precursor was varied into 2, 3 and 4 g, it was obtained that 2 g gave the best photocatalyst properties, namely highest crystallinity, lowest band gap energy, and largest specific surface area. These properties made the 2–CN sample have the best photocatalytic activity (24%). When the amount of the salt melt was varied into 2.5, 5 and 7.5 g, the best photocatalyst properties and photocatalytic activity could be found in the CN–5 sample. Similar to the 2–CN sample, the CN–5 sample gave good crystallinity, suitable low band gap energy, and high specific surface area. This study demonstrated that the amounts of precursor and salt melt are important parameters in synthesizing the crystalline carbon nitride.

<sup>&</sup>lt;sup>b</sup>Specific surface areas were determined via BET technique.

#### **ACKNOWLEDGMENTS**

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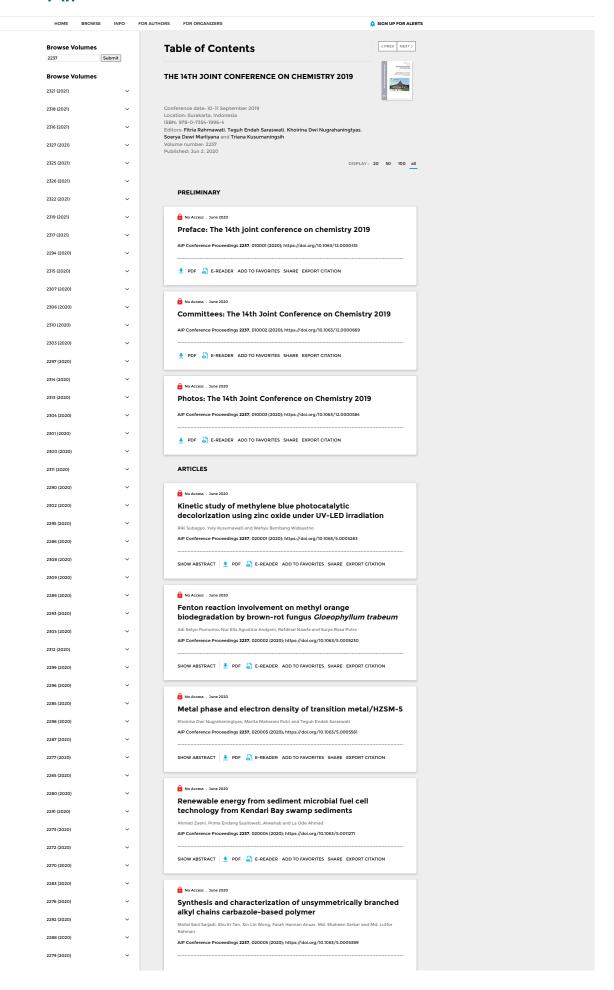
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